



The enhanced near-infrared photocatalytic and photothermal effects of MXene-based heterojunction for rapid bacteria-killing

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ABSTRACT

Bacterial resistance threatens the health of human beings when they combat pathogenic diseases with antibiotics. Herein, an ecofriendly and antibiotics-free 0D/2D Schottky heterojunction ($\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$) was synthesized for the first time to treat bacterial infection with a high efficacy of 99.99% within 20 min under 808 nm NIR light irradiation, which was ascribed to the synergy of enhanced photocatalytic and photothermal effects. The density functional theory calculations show that when a redistribution of charge occurs at the interface between Ag_2S and Ti_3C_2 due to the difference in their Fermi energy levels (E_F), it results in an upward bend of the Ag_2S energy band. Additionally, in the case of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$, the higher conduction band energy of Ag_2S compared to the E_F of Ti_3C_2 and the excellent electrical conductivity of the latter made the photoexcited electrons in Ag_2S , including those hot electrons produced by surface plasma resonance (SPR), rapidly flow to Ti_3C_2 , thus greatly increasing both photocatalytic and photothermal performance of Ti_3C_2 . *In vitro* and *in vivo* tests showed that this photo-responsive system possessed not only highly effective antibacterial efficacy against *Staphylococcus aureus* infection but also excellent biocompatibility simultaneously. Therefore, this hybrid is expected to be a promising ecofriendly platform for the rapid treatment of bacterial infections using NIR light.

1. Introduction

A range of diseases caused by bacterial infections poses serious threats to human health. Each year, approximately 17 million people worldwide die from infectious diseases caused by pathogenic bacteria, and this number is increasing rapidly [1]. Among them, *Staphylococcus aureus* (*S. aureus*) is one of the most common pathogenic bacteria, with roughly 30% of the population colonized by this microorganism [2], which has caused a variety of human diseases such as meningitis, osteomyelitis [3], wound infections, and pneumonia. For a long time, the

discovery of antibiotics largely alleviated the adverse effects caused by bacterial infections [4].

However, the misuse of antibiotics has contributed to the rapid development and spread of bacterial resistance to antibiotics. The emergence of bacterial drug resistance has made once-treatable diseases deadly again, which has been greatly undermining the achievements of modern medicine [5]. Although a new generation of antibiotics can be effective in killing existing drug-resistant bacteria, developing new antibiotics is not only expensive but also time-consuming. In contrast, it takes only a few weeks for bacteria to develop antibiotic resistance. As a

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result, the rate of development of new antibiotics has not kept pace with the rising rate of bacterial resistance [6]. For this reason, there is an urgent need to develop safe and rapid therapeutic strategies as an effective alternative to conventional antibiotics for the treatment of bacterial infections without inducing bacterial resistance.

Currently, many antibiotic-free strategies have been employed to treat bacterial infections through various methods, such as releasing metallic ions [7], electrostatic adsorption, releasing radical oxygen species (ROS) [8], locally increasing temperature [9], and puncturing bacterial membranes by keen-edged nanoneedles [10]. Among them, near-infrared (NIR) light-based phototherapy, including photothermal therapy (PTT) and photodynamic therapy (PDT), is one of the most promising therapeutic strategies due to its deep penetration, biosafety, and rapid and highly effective therapeutic efficacy without bacterial resistance [11,12].

PTT is based on the rapid increase of local temperature, which will denature and inactivate the proteins in the bacteria and lead to bacterial death [13]. PDT is a treatment that excites photosensitizers by light to produce ROS, which can cause severe oxidative damage to bacteria and their final death [14]. Studies have shown that treatment of bacterial infections *in vivo* by applying safe photoinduced heat (less than 55 °C) will not cause damage to healthy tissue [15]. In addition, the local heat generated by the photothermal effect increases the permeability of the bacterial cell membrane, making it easier for antibacterial substances such as reactive oxygen species (ROS) from the environment to enter the bacteria and thus inactivate them. Therefore, a synergistic antibacterial strategy using PTT-enhanced PDT can achieve the effective eradication of bacteria at lower temperatures [16].

Besides its photothermal, excellent biocompatibility and chemical stability, the narrow band gap of 0.9 eV allows Ag₂S to be excited by NIR light for photocatalytic reactions [17–20]; hence, this material has been used for phototherapy [21]. However, the narrow band gap can accelerate the recombination of photogenerated electrons and holes in Ag₂S, thus greatly limiting its photocatalytic reaction [22]. Recently, MXene, a new type of two-dimensional material [23], has attracted much attention and been applied in a variety of fields such as water treatment, catalysis [24,25], intracellular fluorescence imaging, antibacterial [26], and anticancer treatment, due to its unique structure and ideally tunable physicochemical properties together with its good biocompatibility and NIR-responsive properties [27].

Furthermore, the ultra-high electron conductivity of MXene makes it easy for electrons to flow from the semiconductors to its surface, thus ensuring effective electron-hole separation. At the same time, the high specific surface area of MXene provides more reactive sites for photocatalytic reactions [28]. In addition, the abundant hydrophilic groups on the surface of MXene enable it not only to easily establish strong links with various semiconductors, but also to interact strongly with water molecules, which ensures the stable performance of MXene in aqueous solutions [29]. In view of the aforementioned, we propose a hypothesis concerning if a Schottky heterojunction of Ag₂S/Ti₃C₂ can be constructed with enhanced photocatalytic and photothermal performance through a continuous flow of NIR-excited electrons including those hot electrons produced by local surface plasma resonance (LSPR) from Ag₂S to Ti₃C₂ at the interface between them.

In this work, based on the above hypothesis, a novel 0D/2D Schottky heterojunction of Ag₂S/Ti₃C₂ was prepared for the treatment of bacteria-infected wounds by the synergy of PTT and PDT under NIR light irradiation. Ag₂S nanoparticles were grown *in situ* on the surface of two-dimensional Ti₃C₂ using a simple wet chemical method. The structure and photocatalytic mechanism of Ag₂S/Ti₃C₂ heterojunction were investigated in detail by DFT calculations. Subsequent *in vitro* and *in vivo* results showed that the introduction of Ti₃C₂ greatly enhanced the photocatalytic and photothermal performance of Ag₂S, which endowed the hybrid with highly effective antibacterial efficacy under NIR light irradiation within a short time.

2. Experimental section

2.1. Preparation of samples

2.1.1. Pretreatment of MXene

At room temperature, Ti₃AlC₂ (purchased from Nanjing Mingshan New Material Co., Ltd., 99.5%) was dispersed in HF solution to etch Al to prepare Ti₃C₂. Generally, 5 g of Ti₃AlC₂ powder was dispersed in 50 mL of HF aqueous solution (40%) and soaked at room temperature for 24 h. After the etching process was completed, the sample was collected by centrifugation and washed with deionized water and ethanol to obtain Ti₃C₂.

The Ti₃C₂ etched by the HF was dispersed in 50 mL DMSO and magnetically stirred at room temperature for 36 h. After the intercalation process was completed, the sample was collected by centrifugation and washed with deionized water and ethanol, and then the powder was freeze-dried and collected.

2.1.2. Preparation of Ti₃C₂ nanosheets

The intercalated Ti₃C₂ powder was sonicated for layer exfoliation. Generally, 0.5 g of Ti₃C₂ black powder was dispersed in 75 mL of deionized water and sonicated with a high-powered ultrasonic cleaner for 4.5 h. The samples were then collected by centrifugation and washed repeatedly with deionized water and ethanol to obtain pure Ti₃C₂ nanosheets.

2.1.3. Synthesis of Ag₂S/Ti₃C₂ hybrids

10 mL of aqueous solution containing 0.15 g of thiourea (Tu) was prepared, to which 11 drops of ammonia at a concentration of 7% were added. Ti₃C₂ nanosheets were weighed according to the different mass ratios (10%, 20%, and 40%) of Ti₃C₂ nanosheets to Ag₂S, and then dissolved in 10 mL of deionized water and dispersed homogeneously by sonication. We then added 10 mL of uniform Ti₃C₂ dispersion into the TU solution and stirred it magnetically for 15 min. At the same time, 10 mL 0.3 M AgNO₃ aqueous solution was prepared. A 10 mL AgNO₃ aqueous solution was slowly added drop by drop to the above mixed solution to initiate the reaction, and magnetic stirring was conducted at room temperature for 30 min. After the reaction was completed, the complex products were collected by centrifugation (12,000 rpm/min, 10 min), then alternately washed with deionized water and ethanol three times, and finally dried in a vacuum at 60 °C.

2.1.4. Synthesis of pure Ag₂S nanoparticles

10 mL of deionized water was used to replace 10 mL of homogeneous Ti₃C₂ dispersion in the above reaction, and the rest of the synthesis process was kept constant. The successful synthesis of Ag₂S nanoparticles was indicated when the color of the solution changed from colorless to deep black within 30 min after initiating the reaction.

2.2. Characterization

The crystal structure of the samples was measured by X-ray diffraction (XRD, D8A25, Bruker, Germany). The Raman spectra of the materials were measured by inVia Reflex Raman microspectrometer (Renishaw, England). X-ray photoelectron spectroscopy (XPS) was obtained with an instrument (ESCALAB 250 XI, Thermo Scientific, USA). *In situ* and *ex situ* XPS measurements of the materials were detected by X-ray photoelectron spectrometer (ThermoFischer, ESCALAB 250Xi). The nitrogen adsorption and desorption behaviors of materials at 77 K were determined by Brunauer-Emmett-Teller (BET, JW-BK112). The microscopic morphology, structure and crystal structure of the samples were observed with field emission scanning electron microscopy (FE-SEM, ZEISS Sigma 500, Germany) equipped with energy-dispersive spectroscopy (EDS) and a high-resolution transmission electron microscope (HRTEM, Titan G260–300). The absorption spectra of the aqueous solutions of the materials and the absorbance at 808 nm were determined

by the enzyme calibrator (SpectraMax i3) (Molecular Devices). The diffuse reflectance spectrum of the material was obtained by UV-3600 ultraviolet–visible–near infrared (UV–vis–NIR) spectrophotometer (Shimadzu, Japan) measurement. The model number of the instrument used to measure ultraviolet photoelectron spectroscopy (UPS) is PHI5000 VersaProbe III (Scanning ESCA Microprobe) SCA (Spherical Analyzer). During the test, the sample bias was -5 V. The photoluminescence (PL) emission spectra of the materials were obtained by fluorescence spectrophotometer (LS-55, Perkin Elmer, USA) measurements.

2.3. Theoretical calculation details

In our calculation process, all the spin theoretical simulations were carried out on the Vienna Ab-initio Simulation Package (VASP), version 5.4.1 [30]. Generalized gradient approximation (GGA) in the form of Perdew-Burke-Ernzerhof (PBE) [31] function was used to evaluate electron-electron exchanges and related interactions, and a projector enhanced wave (PAW) method was implanted to represent the nuclear-electron (valence electron) interactions effect. The plane-wave basis function set a dynamic cut-off energy of 500 eV. A Monkhorst-Pack meshes [32] with a size of $5 \times 5 \times 5$ was used to sample the bulk Brillouin zone, and $5 \times 5 \times 1$ was for the surface. The geometry of the ground-state atoms was optimized by reducing the force below 0.02 eV/Å and setting the convergence criterion for energy to 1.0×10^{-5} eV/cell. The accuracy of the electronic structure and total energy calculation results was ensured by using the Gaussian method. Van der Waal (vdw) interactions were included in Grimme's DFT-D3 method [33] described to obtain a better description of intermolecular interactions.

2.4. Photocatalytic properties testing of materials

2.4.1. Photoelectrochemical detection

The sample was configured into a homogeneous aqueous dispersion with a concentration of 5 mg/mL, and subsequently 50 μ L of the material solution was added dropwise onto the surface of a 6 mm diameter titanium sheet. After the sample on the surface of the titanium sheet was completely dried, it was used as the working electrode. The reference electrode was the Ag/AgCl electrode and the platinum electrode was the counter electrode. The photoelectrochemical properties of the samples in Na₂SO₄ (0.5 M) electrolyte were measured in a standard three-electrode system using an electrochemical workstation (PGSTAT302N, AUT87820, Netherlands) under 808 nm laser irradiation conditions.

2.4.2. Detection of photocatalytic generation of reactive oxygen species

2,7-Dichlorofluorescein diacetate (DCFH-DA) was used as a reactive oxygen fluorescent probe to detect the total amount of reactive oxygen species produced by the samples under irradiation. Briefly, 180 μ L of DCFH was taken in a 96-well plate, followed by the addition of 20 μ L of aqueous material solution (5 mg/mL) and mixed well. For the Ctrl group, 20 μ L of deionized water was added. Each well was irradiated with an 808 nm laser for 20 min, and the fluorescence intensity of the solution (excitation wavelength 488 nm and absorption wavelengths 525 nm) was recorded every two minutes using an enzyme marker.

The production of ¹O₂ during photocatalysis was detected using 1,3-Diphenylisobenzofuran (DPBF). We added 180 μ L of DPBF DMSO solution (10 μ g/mL) to the 96-well plate, followed by 20 μ L of material aqueous solution (5 mg/mL). All samples were shaken for 10 min under dark conditions to mix well. After irradiation at 808 nm NIR for 20 min, the absorption spectra of the solutions were detected by enzyme standardization.

The electron spin resonance (ESR) spectra of the materials were measured by a JES-FA200 spectrometer (JEOL, Tokyo, Japan). Briefly, ¹O₂ production during Ag₂S/Ti₃C₂-20 photocatalysis was measured by electron spin resonance under 808 nm NIR laser irradiation conditions with 2,2,6,6-tetramethylpiperidine (Aladdin) as a spin trap.

2.5. Photothermal performance testing of materials

2.5.1. Photothermal heating curve and heating thermal image

200 μ L of aqueous solution of the material at a concentration of 500 μ g/mL was taken in a 96-well plate and irradiated with an 808 NIR laser (0.67 W/cm²). The temperature of the samples was recorded and photographed every 60 s with a FLIR thermal camera (FLIR-E50, Estonia).

2.5.2. Photothermal cycle curve

Like the above procedure, the temperature of Ag₂S/Ti₃C₂-20 was recorded every minute under the condition that the 808 NIR laser was turned on for 20 min followed by 20 min off for a photothermal cycle. This process was repeated five times.

2.5.3. Photothermal conversion efficiency calculation

The Ag₂S/Ti₃C₂-20 photothermal conversion efficiency (η) is obtained by the following equation (Eq. (2.1)) [34]:

$$\eta = \frac{hS(T_{\max} - T_0) - Q}{I(1 - 10^{-A})}, \quad (2.1)$$

where h is the heat transfer coefficient, S is the surface area of the vessel, $T_{\max}$ is the maximum equilibrium temperature during heating, T_0 indicates the ambient temperature, Q is the heat absorption rate of the 96-well plate (polystyrene), I represents the 808 NIR laser power, and A is the absorbance of Ag₂S/Ti₃C₂-20 at 808 nm.

The time constant (τ_s) for Ag₂S/Ti₃C₂-20 was calculated by equation (Eq. (2.2)) during the cooling process.

$$t = -\tau_s \ln \theta = -\tau_s \ln \frac{T - T_0}{T_{\max} - T_0} \quad (2.2)$$

At this point, assuming that the thermal input and output of the system are equal, the

$$hS = \frac{\sum_i m_i C_{p,i}}{\tau_s} \approx \frac{m_{H_2O} C_{H_2O}}{\tau_s} \quad (2.3)$$

In formula (Eq. (2.3)), m_{H_2O} and C_{H_2O} are the mass of water and the specific heat capacity of water, respectively.

2.6. In vitro antimicrobial test

The *in vitro* antimicrobial properties of the materials were quantitatively evaluated with bacterial plate coating under photoexcitation. *S. aureus* was cultured in a standard Luria-Bertani (LB) culture medium. A density of 5×10^6 CFU/mL of *S. aureus* solution was added to the 96-well plate with 180 μ L, followed by 20 μ L of material solutions (Ti₃C₂, Ag₂S, Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40) at a concentration of 5 mg/mL to the experimental group and 20 μ L to the Ctrl with sterilized PBS. Each well was irradiated with an 808 nm laser (0.67 W/cm²) for 20 min, during which the temperature was kept at about 55 °C to avoid the effect of high temperature. In the dark group, the material was co-cultured with the bacteria for 20 min under dark conditions. After the completion of the antibacterial treatment, the bacterial solution in the wells was blown evenly and diluted, and the bacterial plate-coating operation was performed on standard LB agar. This was followed by incubation at 37 °C for 24 h. The calculation of antimicrobial rates was performed according to the following equation (Eq. (2.4)):

$$\text{Antibacterial ratio (\%)} = \frac{C - S}{C} \times 100\% \quad (2.4)$$

where C and S stand for the numbers of bacteria (CFUs) in the control and sample groups, respectively.

The experimental procedures for the research on the antibacterial properties of materials against *Escherichia coli* (*E. coli*) are consistent with the above-mentioned antibacterial experimental procedures.

2.7. Antibacterial mechanism testing

2.7.1. Live/dead fluorescent staining of bacteria

After completion of the *in vitro* light/dark antimicrobial operation as described above, the bacterial solution in 96-well plate was discarded and 100 μL of live/dead fluorescent dye (SYTO9 and PI, LIVE/DEAD BacLight Bacterial Viability Kit, Beyotime) was added in. Subsequently, the well plate was incubated at 37 $^{\circ}\text{C}$ for 20 min. We discarded the dye from the well plate and washed gently with sterilized PBS 3 times. The well plate was dried away from light and photographed with an inverted fluorescence microscope (Olympus, IX73, Japan); live bacteria fluoresce in green while dead bacteria fluoresce in red.

2.7.2. Bacterial morphology observation

After completion of the antimicrobial test, the bacterial solution in the well plate was discarded and washed three times with sterilized PBS. Subsequently, 200 μL of 2.5% glutaraldehyde solution was added to each well for 2 h to fix the bacteria. The wells were washed three times with sterilized PBS and then each well was dehydrated by adding 200 μL of different concentrations of ethanol in a gradient (30%, 50%, 70%, 90%, and 100%) for 15 min each time. After the samples were dried, the bacterial morphology was observed by FE-SEM and photographed.

2.8. Statistical analysis

All the quantitative data were evaluated and analyzed as mean values $\pm$ standard deviations and contrasted via one-way analysis of variance (ANOVA). Values of $*P < 0.05$, $**P < 0.01$, $***P < 0.001$, and $****P < 0.0001$ were considered statistically significant.

3. Results and discussion

3.1. Characterization of synthesized $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids

3.1.1. Synthesis of heterojunction

The preparation process of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids is schematically shown in Fig. S1, illustrating that Ag_2S nanoparticles [35] are uniformly distributed on the surface of Ti_3C_2 nanosheets, which was proven by the morphologies of synthesized materials shown in Figs. 1 and S2.

3.1.2. Morphology of materials

As shown in Fig. S2, the synthesized pure Ag_2S showed an irregular nanostructure (Fig. S2A & S2A1) while Ti_3C_2 displayed a smooth lamellar structure (Fig. S2B & S2B1). After a wet chemical reaction, numerous nanoparticles were uniformly distributed on the surface of the lamellar nanosheets (Fig. S2C and S2C1). The HRTEM images of Ag_2S and Ti_3C_2 further confirmed the irregular nanoparticles of synthesized Ag_2S (Fig. 1A) and the nanosheet-like structure of Ti_3C_2 (Fig. 1B). Fig. 1C clearly shows that Ag_2S nanoparticles were evenly distributed on Ti_3C_2 nanosheets. The point-scan energy dispersive spectroscopy (EDS) detection (Fig. 1E) of Fig. 1C shows that it contains elements such as Ti, C, Ag, and S. To get more accurate element content information, the surface scan EDS mapping by FE-SEM showed that the contents of C, S, Ag, and Ti in $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 were 31.88 at.%, 12.71 at.%, 24.56 at.%, and 30.85 at.%, respectively (Fig. S3A & S3B). The HRTEM of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 showed the crystal domains of Ag_2S and Ti_3C_2 and their respective parallel lattice stripes (Fig. 1D). The d-spacing of 2.42 $\text{\AA}$ corresponds with the (013) crystal plane of Ag_2S , while the d-spacing of 2.15 $\text{\AA}$ corresponds with the (105) crystal plane of Ti_3C_2 , which is in

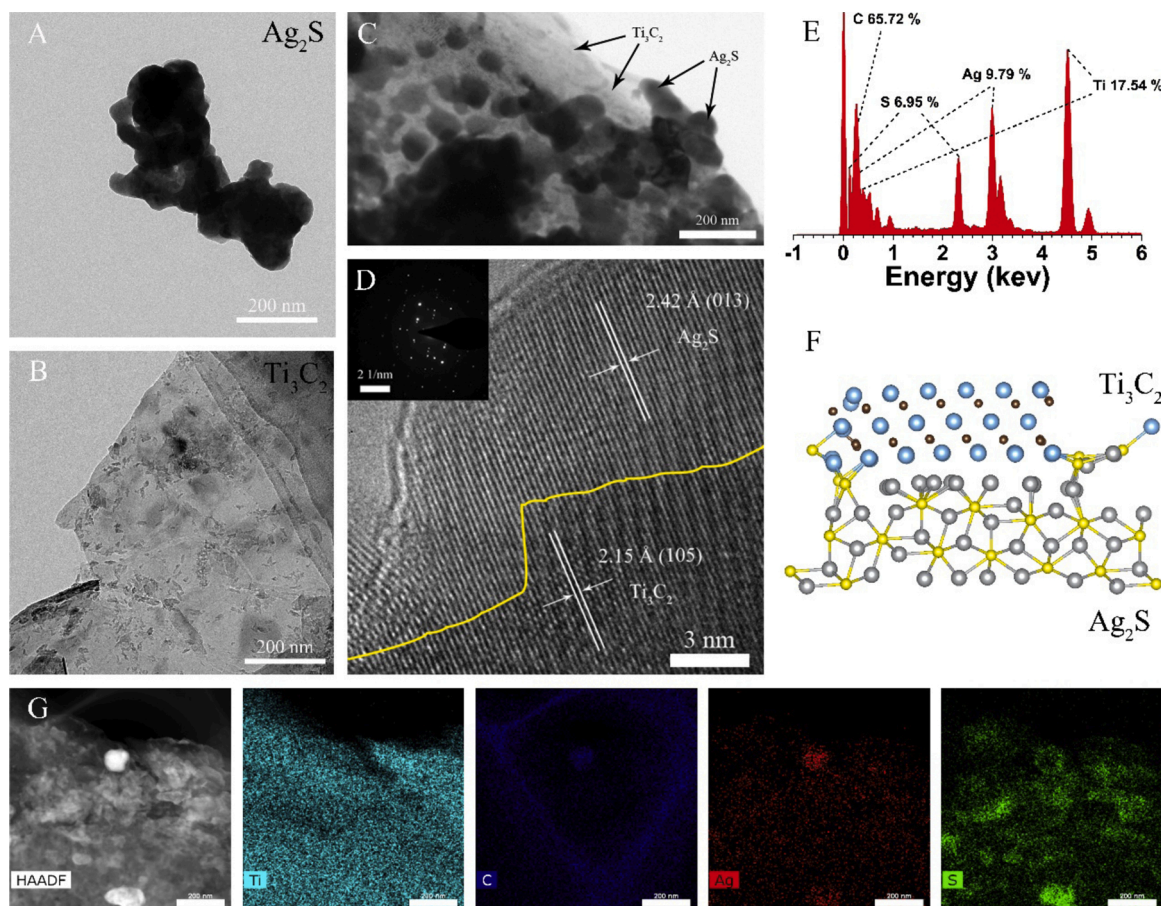


Fig. 1. TEM images of Ag_2S (A), Ti_3C_2 (B), and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 (C). (D) HRTEM image of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 with selected area electron diffraction patterns. (E) EDS detection of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 by TEM. (F) Structural model for theoretical calculations of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$. (G) Elemental mapping of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 by TEM.

accordance with the inserted selected area electron diffraction (SAED) patterns [36]. This result indicated that Ag_2S and Ti_3C_2 were successfully compounded together tightly. Fig. 1F shows the front view of the atomic structure of the Ti_3C_2 layer and the (013) crystal plane of Ag_2S , which will be discussed in detail in the subsequent DFT calculations. The elemental mapping in the TEM image (Fig. 1G) clearly showed the uniform distribution of the elements of Ag, S, Ti, and C in $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20, suggesting the uniform distribution of Ag_2S nanoparticles on Ti_3C_2 nanosheets, which is in agreement with Fig. S3C.

3.1.3. Material component characterization

The X-ray patterns of the synthesized samples are shown in Fig. 2A. The prepared Ag_2S showed obvious peaks of (111), (-112), (120), (-121), (-103), and (200) at 29.0° , 31.5° , 33.6° , 34.4° , 37.7° , and 43.4° , respectively, which corresponds to the characteristic peaks of Ag_2S [22, 35]. The XRD patterns (Fig. S4) of Ti_3C_2 before and after ultrasonic peeling with HF etching and intercalation showed crystalline surfaces at (002), (004), (105), and (110), corresponding to the characteristic peaks of Ti_3C_2 [36,37]. The characteristic peaks of both Ag_2S and Ti_3C_2 still appeared clearly on the X-ray diffractograms of the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrid with a different mass ratio, with different intensity. The above results indicated the successful preparation of Ti_3C_2 , Ag_2S , and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids. Fig. S5 shows the full spectrum of the Raman spectra of the synthesized materials in the range of $50\text{--}2000\text{ cm}^{-1}$. Ag_2S showed no peaks. In contrast, strong Raman signals were detected from both $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids and Ti_3C_2 . The Raman signals of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -10, $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20, and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40 gradually increased with the

increase of Ti_3C_2 content in the hybrids. The enlarged Raman spectra in the range of $250\text{--}800\text{ cm}^{-1}$ showed obvious characteristic peaks of Ti_3C_2 at 414 cm^{-1} and 600 cm^{-1} (Fig. 2B), which is in accordance with previous results [38]. This result further indicates the successful preparation of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids. XPS examination was performed to investigate the functional groups and surface element valence states of Ti_3C_2 and Ag_2S in the hybrids. As shown in Fig. 2C, the survey scan of the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40 shows the existence of the elements of Ag, S, Ti, and C in the hybrid. The narrow scan of Ag 3d displayed that two distinct peaks at 374.2 eV and 368.2 eV (Fig. 2D) are attributed to $\text{Ag } 3d_{3/2}$ and $\text{Ag } 3d_{5/2}$, respectively. In the case of S 2p, two peaks were located at 162.7 eV and 161.5 eV (Fig. 2E), which were assigned to $\text{S } 2p_{1/2}$ and $\text{S } 2p_{3/2}$, respectively. Together with XRD patterns, the above Ag 3d and S 2p signals should be obtained from the synthesized Ag_2S in hybrid [21]. As shown in Fig. 2F, Ti 2p is fitted into three peaks at 455.4 eV , 461.1 eV , and 456.4 eV , indicating that the main forms of Ti in the hybrid are Ti-C (Ti $2p_{3/2}$) and Ti (ii) [36,39,40]. The C 1s narrow scan (Fig. 2G) in the samples can be fitted into C-Ti, C-C, C-O, C=O, and O-C=O, respectively. The high resolution spectra of O 1s (Fig. 2H) can be fitted into two peaks at 531.2 eV and 532.9 eV , corresponding to Ti-OH and H_2O [40].

3.2. Density functional theory (DFT) theoretical calculations

3.2.1. Atomic structure simulation and density state analysis

DFT calculations were performed to study the structural properties and interface interactions between the two phases in the hybrid. During

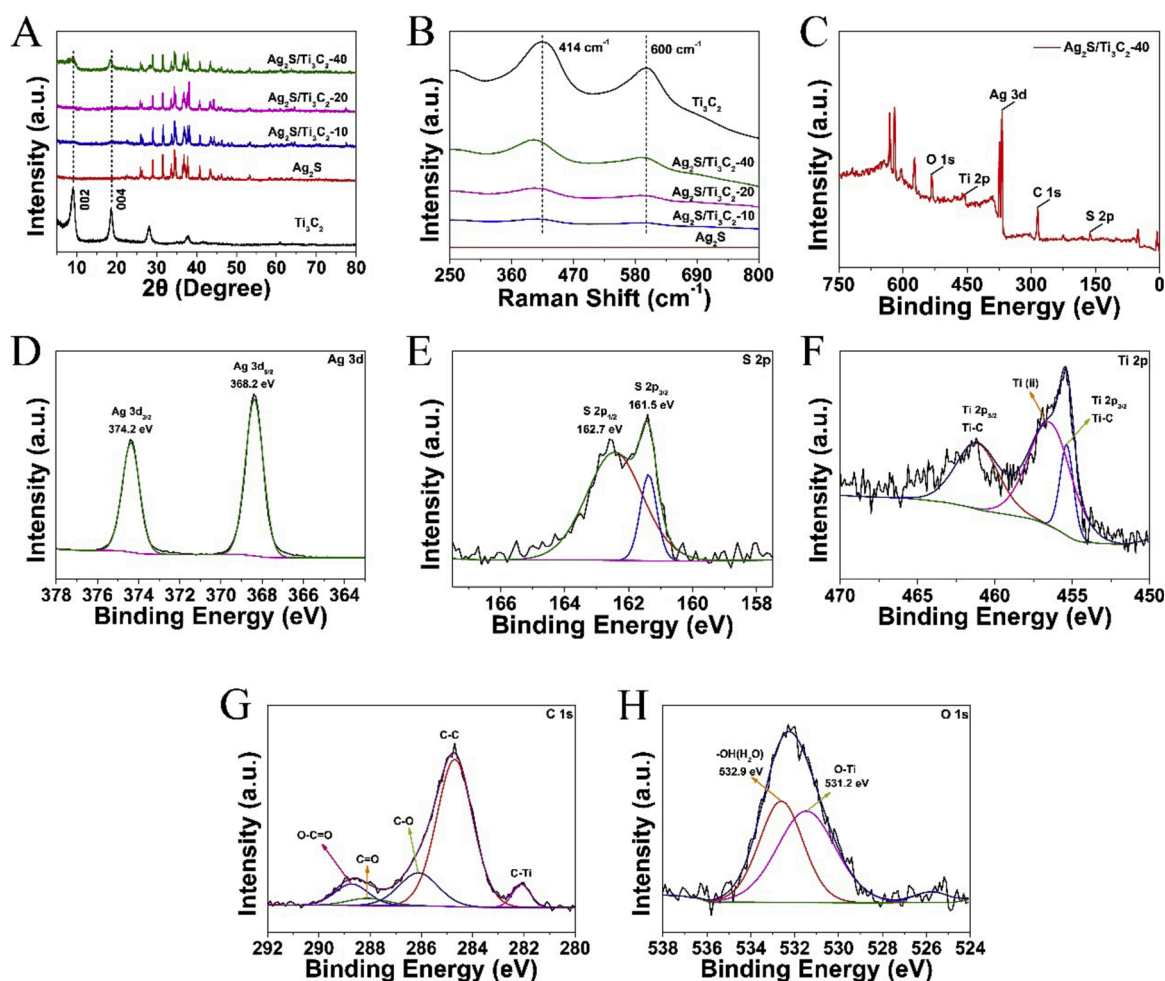


Fig. 2. XRD patterns (A) and Raman spectra (B) of Ti_3C_2 , Ag_2S , $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -10, $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20, and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40. (C) The XPS survey spectrum of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40. High-resolution XPS spectra of Ag 3d (D), S 2p (E), Ti 2p (F), C 1s (G), and O 1s (H) obtained from $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40.

DFT calculation, the atomic structures of the Ti_3C_2 layer and (013) crystal plane of Ag_2S were optimized. As shown in Fig. 3A, the side and top views of Ti_3C_2 show that Ti_3C_2 is a structurally regular laminar structure while Ag_2S is a slab model with a 013 plane. The side view of heterojunction displays that Ti_3C_2 and Ag_2S are tightly bound. To probe the heterojunction photocatalytic mechanism, the density of states (DOS) of Ag_2S , Ti_3C_2 , and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ were calculated to investigate the degree of orbital contribution in the materials, respectively. The Fermi energy level (E_F) is set to 0 eV as a reference. The band gap size of Ag_2S is about 0.8 eV, as shown in the DOS results of Ag_2S (Fig. 3B), which is consistent with the subsequent experimental results. In addition, the valence band (VB) of Ag_2S consists mainly of d-p hybridization of Ag and S, while the conduction band (CB) of Ag_2S consists mainly of s-p hybridization of Ag and S. Fig. 3C shows that the VB of Ti_3C_2 consists mainly of the d-p hybridization of Ti and C, while the CB consists mainly of Ti-d. The CB and VB of Ti_3C_2 partially overlap each other, indicating the good electrical conductivity of Ti_3C_2 . The DOS results of the heterojunction (Fig. 3D) showed that the VB of the Ag_2S (013)/ Ti_3C_2 layer complex is mainly derived from the Ag-d of Ag_2S , while the CB is mainly derived from the Ti-d of Ti_3C_2 . Moreover, the CB and VB of the heterojunction are partially overlapped together, suggesting that the conductivity of the hybrid is much better than that of Ag_2S . Therefore, the electrons of the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ heterojunction will jump from Ag-d of Ag_2S to Ti-d of Ti_3C_2 under 808 nm NIR light irradiation. These results indicate that during the photocatalytic process, the photogenerated electrons in Ag_2S are rapidly guided away by Ti_3C_2 and thus flow to the surface of Ti_3C_2 , which extends the lifetime of the photogenerated electrons, thus improving the photocatalytic performance of the

heterojunction [41].

3.2.2. Electrostatic potential and built-in electric field simulation

To further study the interface charge transfer path and formation mechanism of the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ heterojunctions, the electronic properties of Ti_3C_2 , Ag_2S , and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ heterojunctions were evaluated by DFT calculation. As shown in Fig. 4A(A1) and B(B1), the calculated work function (WF) of the Ag_2S (013) plane and Ti_3C_2 layer is 4.45 eV and 4.00 eV, respectively, revealing the corresponding calculated Fermi levels of -1.12 eV and -1.45 eV versus vacuum level, respectively; i.e., the Fermi level of Ti_3C_2 is lower than that of Ag_2S . Therefore, under dark conditions, when Ti_3C_2 and Ag_2S are in close contact, electrons on the surface of Ag_2S can flow into the surface of Ti_3C_2 through the interface. This result can be directly confirmed by the differences in plane average electron density (Fig. 4C) and the difference in three-dimensional charge density (Fig. 4C1), suggesting that there is a redistribution of charge at the interface of the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ heterojunction. In the constructed $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ heterojunction model, the blue and yellow regions represent the electron depletion zone and the electron accumulation zone, respectively, and the difference in plane average electron density ($\Delta\rho$) also reveals the accumulation of electrons. The above calculation results show that once the Ag_2S nanoparticle contacts the Ti_3C_2 nanosheets tightly, a built-in electric field can be formed spontaneously at the interface between the two phases, inducing the formation of a Schottky heterojunction [42].

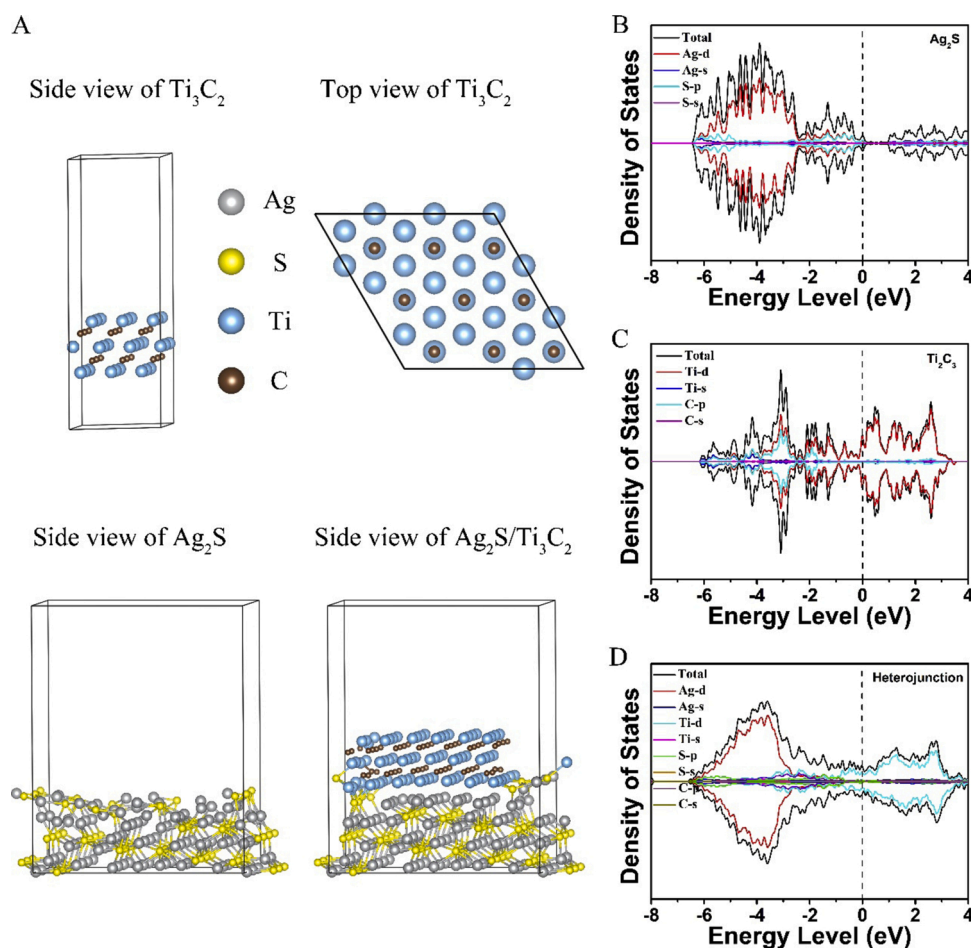


Fig. 3. (A) Structure model of Ti_3C_2 , Ag_2S , and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ of different perspectives after structure optimization. Theoretical calculations of DOS of Ag_2S (B), Ti_3C_2 (C), and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ (D), respectively.

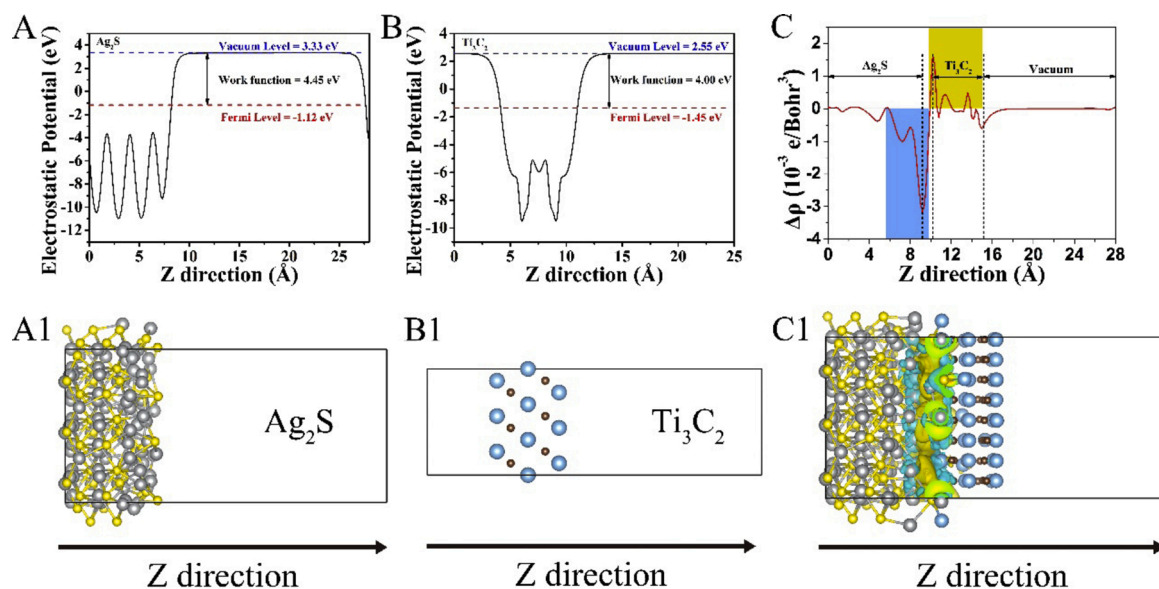


Fig. 4. (A) The calculated work function and (A1) corresponding structural model of (013) plane of Ag₂S. (B) The calculated work function and (B1) corresponding structural model of Ti₃C₂. (C) The difference in plane average electron density ($\Delta\rho$) and the difference in three-dimensional charge density (C1) of Ag₂S/Ti₃C₂ heterojunction.

3.3. Photocatalytic performance evaluation

3.3.1. Band structure detection

Fig. S6A shows the UV–vis diffuse reflectance spectra of Ag₂S, Ti₃C₂, and Ag₂S/Ti₃C₂ hybrids. Since Ag₂S is a typical indirect semiconductor, the band gap energy relationship diagram of the sample (Fig. S6B–E) can be obtained by calculation based on the UV–vis spectra in Fig. S6A according to the Kubelka-Munk formula [17,43]. The calculated band gap of Ag₂S, Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 is 0.81, 0.77, 0.47, and 0.44 eV, respectively, suggesting the decreased band gap of the samples with the increase of Ti₃C₂ content in the hybrid, and thus resulting in the red-shift of absorption light. Therefore, increasing the content of Ti₃C₂ in the hybrid can increase the light absorption (Fig. S6F).

As shown in Fig. 5A, in the range of 360–1000 nm, the absorption intensity of pure Ti₃C₂ is the strongest, and pure Ag₂S shows the weakest absorption intensity. Meanwhile, the absorption intensities of Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 gradually increased, which is consistent with the above-mentioned results. That is to say, the introduction of Ti₃C₂ enhances the light-trapping ability of the hybrid heterojunction.

According to the ultraviolet photoelectron spectroscopy (UPS) spectra (Fig. 5B–D), the calculated work functions (Φ) of Ag₂S, Ti₃C₂, and Ag₂S/Ti₃C₂-20 are 3.9, 4.07, and 4.18 eV, respectively, based on the energy-level information derived from the UPS measurement (Fig. S7) and previous literature [44]. The energy difference between E_F and valence band (E_{VB}) comes from the low binding energy tail (E_{edge}) [45]. Thus, the E_{VB} and conduction band (E_{CB}) of Ag₂S are -4.47 eV and -3.66 eV, respectively. The E_{VB} and E_{CB} of the sample (Ag₂S/Ti₃C₂-20) after the combination of Ag₂S and Ti₃C₂ are -4.9 eV and -4.09 eV, respectively.

3.3.2. Photoelectrochemical detection

The photoelectrochemical properties of Ag₂S, Ti₃C₂, and Ag₂S/Ti₃C₂ hybrids were investigated and are shown in Fig. 5E. It is evident that the photocurrent intensity of the Ag₂S/Ti₃C₂ hybrids group is significantly higher than that of Ag₂S and Ti₃C₂, indicating that the combination of Ag₂S and Ti₃C₂ can effectively promote the separation of photogenerated electron-hole pairs and generate photogenerated charge carriers under NIR light irradiation. Among the Ag₂S/Ti₃C₂ hybrids, Ag₂S/

Ti₃C₂-20 has the highest photocurrent intensity, followed by Ag₂S/Ti₃C₂-10 and Ag₂S/Ti₃C₂-40 has the relatively lowest. The electrochemical impedance spectroscopy (EIS) measurements (Fig. 5F) showed that the decreased slope of the curves as the Ti₃C₂ content in the heterojunction increased, suggesting that the increase of Ti₃C₂ content reduced the charge transfer resistance of the heterojunction and thus accelerated the charge transfer rate, which provided more charge carriers for the following photocatalytic reactions. Since the recombination of photogenerated electron-holes will emit energy in the form of fluorescence, the higher photoluminescence (PL) intensity indicates the recombination of more photogenerated electron-hole pairs. Fig. 5G shows that higher Ti₃C₂ content in the heterojunction results in much lower PL intensity, indicating that the addition of Ti₃C₂ can be beneficial for separating photogenerated-electrons in the hybrid and thus enhancing the photocatalytic activity of the material. This result is in agreement with the above-measured photocurrent density.

3.3.3. Detection of reactive oxygen species produced by photocatalysis

As an ROS-sensitive sensor, 2,7-dichlorofluorescein diacetate (DCFH-DA) can sensitively detect the presence of ROS in the surrounding environment [46]. After being irradiated by 808 nm NIR light for 20 min, the results of the DCFH-DA assay (Fig. 5H) showed that the total ROS content in the Control (Ctrl), Ti₃C₂, Ag₂S, Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 groups changed by 4.06, 4.48, 35.87, 93.33, 101.33, and 92.59-fold, respectively. The Ti₃C₂ group showed a very low signal of almost the same as the control group, indicating no ROS production during the light exposure. Meanwhile, the Ag₂S group also exhibited little ROS yields during the light exposure. In contrast, all the hybrids showed much higher ROS yields, suggesting that the addition of Ti₃C₂ greatly enhances the photocatalytic performance of Ag₂S. Among the three hybrid groups, Ag₂S/Ti₃C₂-40 showed the lowest ROS yields with 20 min NIR light irradiation, which was attributed to the lowest content of Ag₂S in Ag₂S/Ti₃C₂-40. In addition, the ROS of the Ag₂S/Ti₃C₂-10 and Ag₂S/Ti₃C₂-20 groups showed the same trend within the first 4 min of light exposure, and the total amount of ROS generated by the Ag₂S/Ti₃C₂-20 group was slightly higher than that by the Ag₂S/Ti₃C₂-10 group in 20 min, suggesting that 20% (mass ratio of Ti₃C₂:Ag₂S) has the strongest photocatalytic performance under 808 nm NIR light excitation. The species of generated ROS were further determined by the photodegradation of 1,3-diphenylisobenzofuran (DPBF)

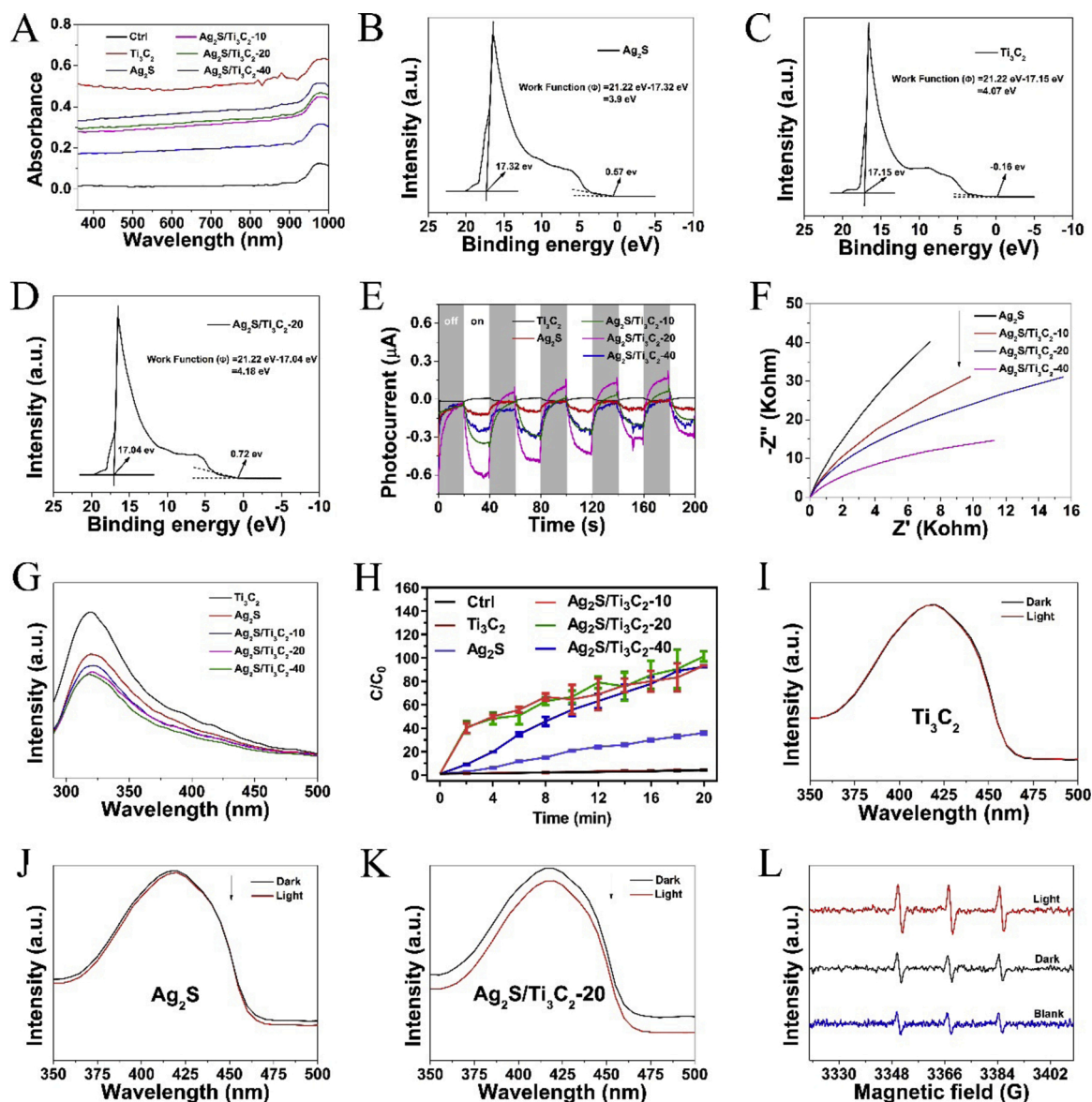


Fig. 5. (A) Absorption spectra of aqueous solutions of materials (500 $\mu\text{g/mL}$). UPS spectra of Ag_2S (B), Ti_3C_2 (C), and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 (D). Photoelectrochemical measurements including photocurrent responses (E) and EIS tests (F) of samples. (G) PL spectra of materials. (H) ROS production of samples detected by DCFH under 808 nm NIR irradiation. Detection of reactive oxygen species in photocatalytic processes: Degradation curves of DPBF of Ti_3C_2 (I), Ag_2S (J), and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 (K); (L) ESR measurements of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20.

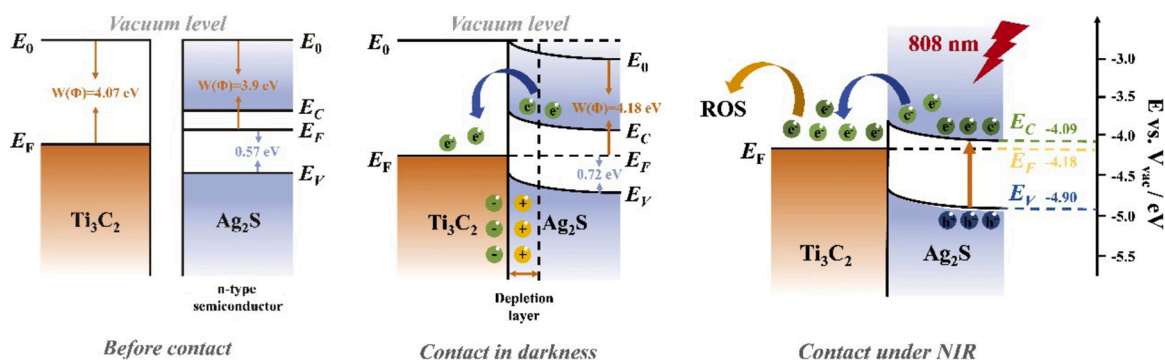


Fig. 6. Schematic diagram of the changes in the energy band structure and internal electric field near the interface of the two phases under different conditions (before contact, contact in darkness and contact under NIR) for Ti_3C_2 and Ag_2S , and the specific process of ROS production during photocatalysis.

and electron spin resonance (ESR) [47]. There was no decrease in the absorption peak around 415 nm in the Ti_3C_2 group before and after irradiation with 808 nm NIR light (Fig. 5D), while there was a slight decrease in the Ag_2S group (Fig. 5J). In contrast, a significant decrease was observed in the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 group (Fig. 5K). In addition, an ESR assay with 2,2,6,6-tetramethylpiperidine (TEMP) as a trapping agent showed the enhanced 1:1:1 signal in $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 after light exposure (Fig. 5L). Both DPBF and ESR results showed that $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 produced $^1\text{O}_2$ under 808 nm NIR light excitation. Overall, the above results indicate that $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ hybrids have good photocatalytic activity under 808 nm NIR light irradiation, and the $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 group has the strongest photocatalytic performance.

3.4. Photocatalytic mechanism of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$

According to theoretical calculations and photocatalytic performance characterization results, the energy band structure diagram of Ti_3C_2 and Ag_2S in the photocatalytic process is shown in Fig. 6. Before

Ti_3C_2 and Ag_2S contact, the calculated WF of Ti_3C_2 (4.07 eV) was greater than that of Ag_2S (3.9 eV) (Fig. 5B and C). After tightly contacting each other, the electrons tended to migrate from the side with a small WF to the surface of a material with a larger WF. Therefore, when Ag_2S contacted Ti_3C_2 under dark conditions, the electrons on the surface of the former flowed to the latter until the two Fermi levels reached a balance. In the equilibrium state, a Helmholtz electric double-layer structure was formed at the interface of Ti_3C_2 and Ag_2S . In the Helmholtz double-layer structure, Ti_3C_2 was negatively charged, and an electron-depletion layer formed near the surface of Ag_2S and was positively charged. Above theoretical calculations have confirmed that a built-in electric field is generated at the interface due to the redistribution of charges when the two phases are in contact (Fig. 4). According to the UPS calculation results, Ag_2S is a typical n-type semiconductor because its Fermi level is closer to CB. When the electrons in Ag_2S are repelled by the negatively charged Helmholtz layer in Ti_3C_2 , the potential energy of the semiconductor rises, which eventually causes the energy band to bend upward [48].

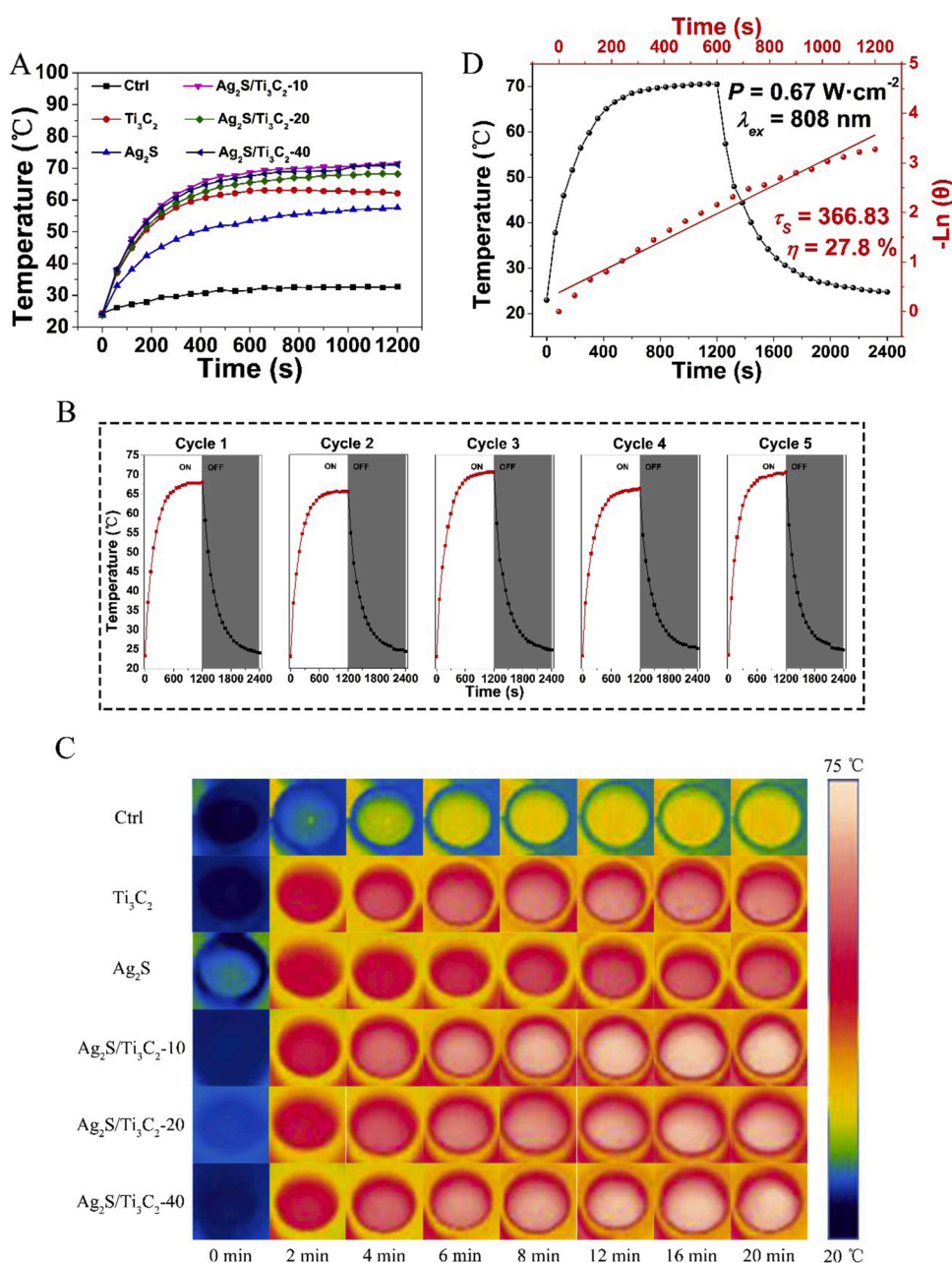


Fig. 7. The photothermal performance test of materials under 808 nm NIR ($0.67 \text{ W}/\text{cm}^2$) irradiation conditions. (A) The photothermal heating curves of Ti_3C_2 , Ag_2S , $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -10, $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20, and $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -40. (B) Photothermal cycle curves (5 cycles) of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20. (C) Thermal images corresponding to the temperatures at different time points during the warming of the material in Fig. 7A. (D) Measurement and calculation of the photothermal conversion efficiency of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20 under this irradiation condition.

Under 808 nm NIR light irradiation, Ag₂S was excited. Subsequently, the electron-hole pairs on the VB of Ag₂S separated, and the photo-generated electrons moved from VB to CB. Because of the bent-upward energy band of Ag₂S and its higher CB than E_F of Ti₃C₂, the photo-generated electrons on the CB of Ag₂S are more inclined to flow to the surface of Ti₃C₂ and be guided away rapidly, resulting in the effective separation of photogenerated electron-hole pairs. The separated photogenerated electrons and holes participate in the redox reaction in the environment to generate a large amount of ROS.

3.5. Photothermal performance

3.5.1. Exploration of photothermal performance

The photothermal properties of the materials were investigated under 808 nm NIR light irradiation by using the device diagram as shown in Fig. S10, where the concentration of the aqueous solution of the materials was kept the same as that used in the antibacterial experiments. Fig. 7A shows the photothermal temperature rise curve of the material. After 20 min of illumination, the temperature of the Ctrl, Ti₃C₂, and Ag₂S groups increased from room temperature (around 24.1 °C) to 32.7 °C, 62.1 °C, and 57.6 °C, respectively. The Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 groups increased from room temperature to 71.5 °C, 68.2 °C, and 71.1 °C, respectively. Fig. 7C shows the thermal images at different time points during the temperature rise of the material in Fig. 7A. The photothermal temperature rise curves and their corresponding thermograms can be used to visualize that the Ag₂S/Ti₃C₂ hybrids formed by the combination of Ag₂S and Ti₃C₂ have better photothermal conversion performance compared with those before the combination. We have measured five sets of photothermal ramp-up/down cycling curves for Ag₂S/Ti₃C₂-20, and the results show (Fig. 7B) that the heterojunction has good photothermal switching response performance and photothermal stability [49]. In addition, the photothermal conversion efficiency of Ag₂S/Ti₃C₂-20 ($\eta = 27.8\%$) was calculated from the time constant ($\tau_s = 366.83$) and the maximum steady-state temperature, as shown in Fig. 7D [11]. This further demonstrates the ability of the heterojunction to convert 808 nm NIR light into thermal energy quickly and efficiently.

3.5.2. Photothermal mechanism

Under 808 nm NIR irradiation, the LSPR effect produced by Ti₃C₂ shows only lower photothermal performance [50]. The photogenerated hot electrons and holes of silver sulfide readily compound non-radiometrically in its deep defects to produce low-density phonons. The emitted phonons transfer the energy of the carriers to the lattice and generate heat by achieving lattice vibrations [51]. Based on the above discussion, the band gap of the composite decreases after the composite of Ag₂S and Ti₃C₂. As a result, the composites can generate more hot electrons under light excitation. These hot electrons not only heat up the lattice by generating a higher density of phonons in the deep defects of Ag₂S, but also transfer more to the Ti₃C₂ surface and generate more heat by enhancing the LSPR effect of Ti₃C₂. Therefore, the photothermal properties of the composites are effectively enhanced by the formation of heterojunctions between Ti₃C₂ and Ag₂S.

3.6. In vitro antibacterial activity assay

3.6.1. Antibacterial rate of materials

The antibacterial properties of the material solution (500 µg/mL) were tested against *S. aureus*. During the 808 nm laser irradiation for 20 min, due to the excellent photothermal effect of the material, we ensured that the temperature of the experimental group did not exceed 55 °C throughout to avoid the effect of excessive temperature on the antibacterial results and also to ensure safety (minimal invasion) and effective bacterial destruction [15]. Fig. 8A shows the antimicrobial plate-coating photos of the materials. The results (Fig. 8B) show that there is no significant difference in the number of colonies between Ctrl

and material groups under dark conditions. This phenomenon clearly indicates that all samples did not have antimicrobial activity under dark conditions and reflects the good biocompatibility of the material itself. However, the antibacterial rates of Ti₃C₂, Ag₂S, Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 under light conditions were 76.71%, 62.48%, 97.54%, 99.99%, and 94.78%, respectively, as shown by the plate-coating photos and Fig. 8C. This result indicates that the photo-excited antibacterial performance of Ti₃C₂ and Ag₂S forming heterojunctions (Ag₂S/Ti₃C₂ hybrids) was greatly enhanced. Among the Ag₂S/Ti₃C₂ hybrids heterojunctions, the Ag₂S/Ti₃C₂-20 group had the most excellent and fastest bactericidal performance.

3.6.2. Antibacterial mechanism

To investigate the mechanism of the photoexcitation of materials for antimicrobial activity, we performed bacterial live-dead fluorescence staining and bacterial SEM tests on different samples under light and dark conditions, respectively. The green fluorescence in the live-dead fluorescence graph (Fig. 8D) was for live bacteria, while the red fluorescence was for dead bacteria. The results of the live-dead fluorescence staining showed that both the Ctrl and experimental groups fluoresced green under dark conditions, which indicated that the samples had no antibacterial ability under dark conditions. The fluorescence results of the Ctrl group under light conditions showed that *S. aureus* survived even under irradiation. However, the sample groups under light conditions showed different degrees of red fluorescence, with the Ti₃C₂ and Ag₂S groups showing smaller red fluorescence signals, while the Ag₂S/Ti₃C₂-10, Ag₂S/Ti₃C₂-20, and Ag₂S/Ti₃C₂-40 groups showed stronger red fluorescence signals. The bacterial live-dead fluorescence results remained consistent with the antibacterial results in Fig. 8A, B, and C. Again, this indicates that the antibacterial performance was greatly enhanced by the formation of heterojunctions (Ag₂S/Ti₃C₂ hybrids) between Ti₃C₂ and Ag₂S. Fig. 8E shows the FE-SEM images of *S. aureus*. The results show that the bacteria in the Ctrl and experimental groups were spherical and had smooth surfaces without defects under dark conditions, which indicates that the bacteria survived well in this condition. Under the light irradiation conditions, the surfaces of the bacteria in the experimental groups became rough and had different degrees of depression and breakage, except for the Ctrl group. This phenomenon indicated that the prepared Ag₂S/Ti₃C₂ hybrids effectively killed *S. aureus* under light irradiation. The Ag₂S/Ti₃C₂ hybrids were demonstrated to have good photothermal and photocatalytic properties. When the material was irradiated with 808 nm near-infrared light, the material PDT generated a large amount of ROS, while the excellent PTT of the material increased the ambient temperature of the bacteria. High temperatures caused irreversible bacterial destruction by denaturing the proteins of the bacteria. Also, high temperature levels increased the permeability of the bacterial membrane, which in turn allowed the ¹O₂ produced during photocatalysis to proceed more easily into the bacteria. It caused the death of bacteria by damaging their cell membranes, proteins, and even DNA [52].

The wound healing tests showed that this hybrid system not only possessed highly effective antibacterial efficacy against *in vivo S. aureus* infection, but also had excellent biosafety, which were shown in Fig. S12-S15. The detailed experimental procedures, results and discussion of *in vivo* antibacterial tests and biosafety evaluation are shown in Supporting Information.

4. Conclusion

In summary, we have successfully prepared a novel 0D/2D Schottky heterojunction that has not been reported before by growing Ag₂S nanoparticles *in situ* on the surface of Ti₃C₂ using a simple wet chemical method. The structure optimization, DOS analysis, and charge-density calculation at the two-phase interface of the heterojunction were performed to elucidate in detail the electron transfer paths of the Ag₂S/Ti₃C₂ Schottky heterojunction in the photocatalytic process. The results

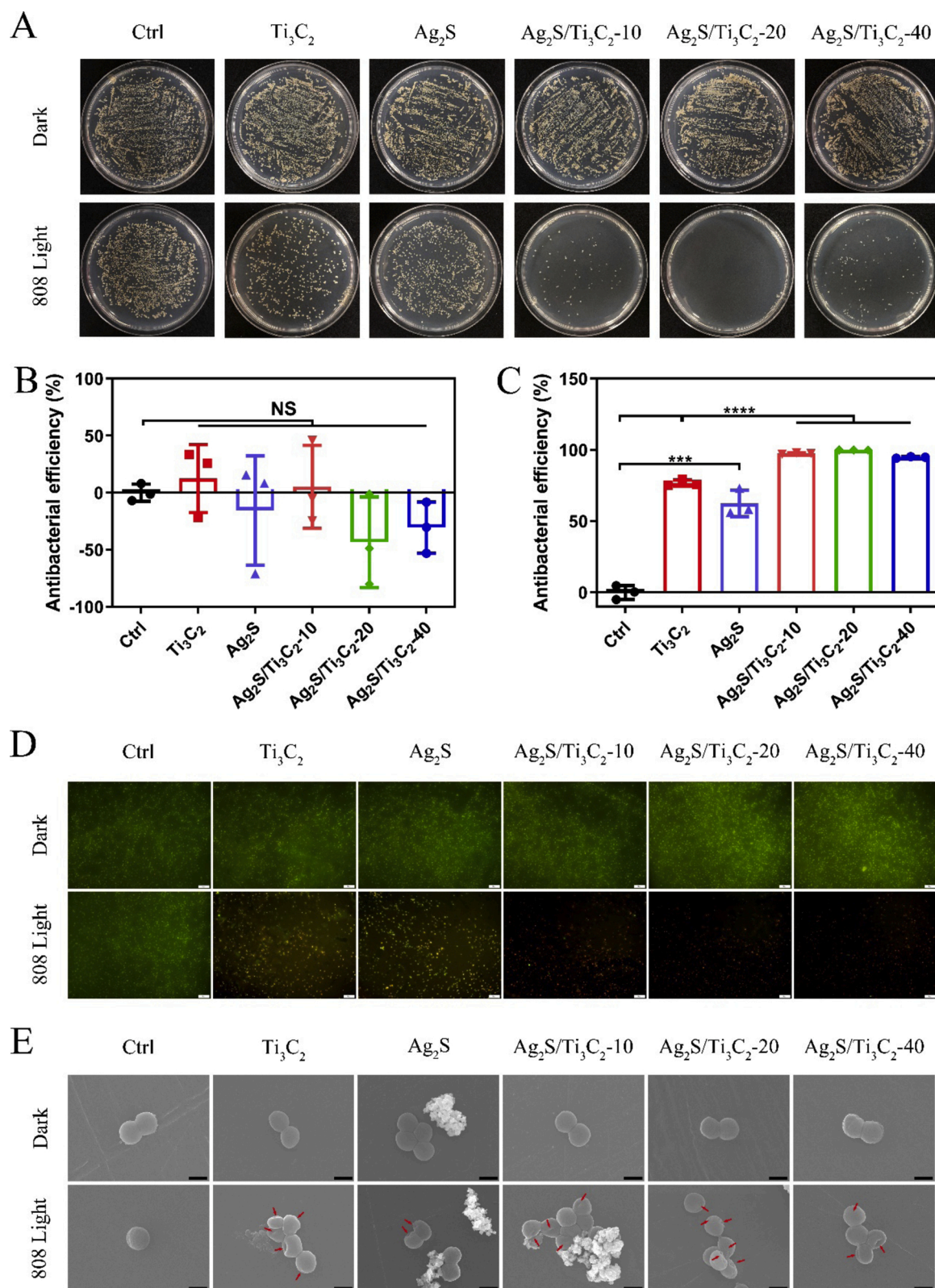


Fig. 8. *In vitro* antimicrobial performance and antimicrobial mechanism testing of materials. (A) Spread plate assay of *S. aureus* with samples after irradiation for 20 min of 808 nm NIR. Corresponding to the antibacterial rate counts under dark conditions (B) and 808 nm NIR irradiation conditions (C) in Fig. 8A. (D) Live-dead fluorescence images of *S. aureus*, live bacterium were stained green, and dead bacterium were stained red. Scale bar = 50 μ m. (E) FE-SEM morphology of *S. aureus*, scale bar = 500 nm. Error bars indicate means $\pm$ standard deviations ($n = 3$): *** $P < 0.001$, **** $P < 0.0001$. NS, not significant ($P > 0.05$).

show that the introduction of Ti_3C_2 enhances the separation efficiency of the photogenerated charge carriers of the complexes and effectively transfers the photogenerated electrons. Therefore, the photocatalytic performance of $\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ Schottky heterojunctions and the ability to produce reactive oxygen species were enhanced. In addition, the heterojunction had an excellent photothermal effect, and the antimicrobial rate of PTT and PDT synergy of the heterojunction ($\text{Ag}_2\text{S}/\text{Ti}_3\text{C}_2$ -20) was 99.99% under 808 nm NIR irradiation for 20 min. It also showed good therapeutic effects in *in vivo* antimicrobial tests. In addition, the material showed good biocompatibility *in vivo* and *in vitro*. These characteristics make this novel 0D/2D Schottky heterojunction promising as a highly efficient platform for the rapid treatment of bacterial infections with NIR. Most importantly, the present work provides a reliable way to explain the detailed mechanism in the heterojunction photocatalytic process and a scheme to optimize the photocatalytic activity of semiconductor photocatalysts.

CRedit authorship contribution statement

Qian Wu: Conceptualization, Methodology, Writing - original draft, Data curation, Investigation. **Lei Tan:** Methodology. **Xiangmei Liu:** Conceptualization, Writing - review & editing, Supervision, Project administration. **Zhaoyang Li:** Methodology. **Yu Zhang:** Conceptualization, Visualization, Writing - review & editing. **Yufeng Zheng:** Conceptualization, Writing - review & editing. **Yanqin Liang:** Methodology. **Zhenduo Cui:** Visualization. **Shengli Zhu:** Methodology. **Shuilin Wu:** Conceptualization, Writing - review & editing, Project administration.

Declaration of Competing Interest

The authors report no declarations of interest.

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Appendix B. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.apcatb.2021.120500>.

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